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EFFECT OF GRINDING FINENESS OF FLY ASH ON THE PROPERTIES OF GEOPOLYMER FOAM

Present paper deals with the development of geopolymer foam prepared from ground F class power station fly ash. The effect of the fly ash fineness on the rheology of the geopolymer paste and the foam properties have been investigated. The raw fly ash was ground in a ball mill for various duration, 5, 10, 20, 30, 60 and 120 min. Geopolymer paste was prepared from the raw and ground fly ash with NaOH – sodium silicate mixture as alkaline activator. Geopolymer foam production was made using H_2O_2 as foaming agent. Additionally, the geopolymer material structure was investigated by Fourier transform infrared spectrometer, the foam cell structure was monitored using optical microscopy. The rheological behaviour of the geopolymer paste changed due to the grinding of fly ash (from Bingham plastic to Newtonian liquid). Grinding of fly ash has a significant effect on the physical properties as well as on the cell structure of the geopolymer foam.

Keywords: fly ash, grinding, rheology, geopolymer foam, cell structure

1. Introduction

Geopolymers are amorphous alumino-silicate materials formed due to reaction between alumino-silicates and oxides with alkaline media (NaOH, KOH solution). Geopolymers can be synthesized at ambient or elevated temperature by alkaline activation of alumino-silicates obtained from mineral materials (perlite), industrial waste (fly ash, slag or red mud) and calcined clay (metakaolin) [1-5].

Alkali activation of these materials together with chemical foaming led to the synthesis of inorganic foam. As foaming agent hydrogen-peroxide [1,6,10] aluminum powder [7-10] or sodium-perborate [6] can be used. The cellular structure of geopolymers can be produced by introducing a large volume fraction of air bubbles into the paste. In this case for the stabilization of the wet foam organic monomers can be used such as detergents, resin soap, glue resins, saponin, and hydrolyzed proteins [10,11].

Geopolymer foam (GPF) possess good physico-chemical and mechanical properties like low density, relative high strength, fire and chemical resistance, thermal stability, good thermal insulation properties [6-9].

The hydrogen-peroxide is thermodynamically unstable and therefore can be easily decomposed to water and oxygen gas with the latter playing the role of the geopolymer paste foaming agent [1,10].

Based on literature [7,9] the geopolymer foam stability of the pore structure at the fresh state strongly depends on the

viscosity of the initial mix. The viscosity can be easily changed by the amount of added liquid activator, i.e. by the liquid/solid ratio. However, rheology of suspensions can be controlled by the size reduction of the solid component, by grinding.

The main aim of the research is to optimize the grinding fineness in order to produce a homogeneous cell structure with appropriate mechanical properties i.e. to investigate the effect of the fly ash fineness on the rheology of the geopolymer paste and the foam properties (pore size distribution, porosity, strength, material structure, cell structure of the geopolymer foam).

2. Materials and methods

Landfilled brown coal fly ash collected from Tiszaújváros (Hungary) was used as main component for geopolymer foam production. Moisture content was 21.49%. Characteristic particle size values of raw fly ash were $x_{50} = 78.04 \, \mu m$; $x_{80} = 145.65 \, \mu m$. The specific surface area (SSA) of raw fly ash was found to be 703.8 cm²/g. The main oxidic components of raw fly ash were following: SiO₂ (60.9%), Al₂O₃ (27.1%), Fe₂O₃ (4.98%), CaO (1.55%), MgO (0.94%) and SO₃ (0.068%).

The raw fly ash was mechanically activated for various duration (5, 10, 20, 30, 60 and 120 min) in a tumbling laboratory ball mill with the size of Ø305×305 mm. During the grinding the specific grinding energy was measured. The particle size distribution of the raw and ground fly ashes had been investigated by Horiba LA950 V2 type laser particle size analyzer, "outer"

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specific surface area was calculated from distribution data by software using shape factor 1.

The geopolymer foam production shows Fig. 1. Waterglass and 12 M NaOH solution were used as alkaline activator. The liquid/solid ratio (L/S) of geopolymer paste was 0.67. Hydrogen peroxide (30 mass % solution) was used as chemical foaming agent. Amount of $\rm H_2O_2$ solution was 3 w/w% in fly ash mass. The heat curing was carried out at 50°C for 10 hours. Three specimens of geopolymer foam were made in all cases for the strength investigation. The uniaxial compressive strength was measured at the age of 7 days. In addition specimen density, cell structure and pore size of geopolymer foam and viscosity of the geopolymer paste were examined.

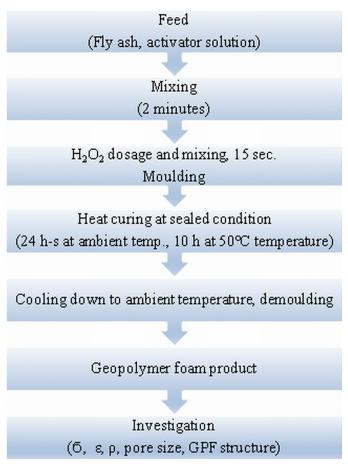


Fig. 1. Geopolymer foam production

In order to measure the rheological properties of the geopolymer paste, it had to be diluted to L/S ratio of 0.82. Denser paste could not be measured due to the high viscosity of the paste. The measurements were carried out with cylindrical geometry and flat geometry measuring head where cylinders were not able to use.

The cell size and shape in geopolymer foam was monitored using C. Zeiss Discovery V.12 optical microscope. The material structure of the geopolymer foam products was investigated by JASCO FT-IR 4200 type Fourier Transformed Infrared Spectrometer in reflection mode, using a diamond ATR.

3. Results and discussion

Effect of grinding on the fly ash fineness

Fig. 2 shows the produced specific surface area and median particle size (x_{50}) as function of specific grinding energy. Based on this figure it can be stated, that SSA increasing almost linearly as function of specific grinding energy. After 120 minutes grinding in a ball mill 10.79 μ m median particle size was achieved from the initial value of 78.04 μ m. In addition the "outer" specific surface area increase was significant; from 703.8 cm²/g to 5876.8 cm²/g using 368 kJ/kg specific energy input.

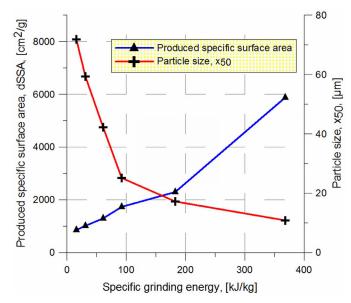


Fig. 2. Grinding kinetics of fly ash

Effect of fly ash fineness on the geopolymer paste viscosity

The examined sample of fly ash and activator mixture showed changing rheological behaviour according to the fineness of the fly ash powder (Fig. 3). The rheology of the mixture of the activator and the raw fly ash and those were taken after 5 minutes grinding time were not possible to measure in the actual device has been being used due to particles were too coarse to fit into the gap of the plate geometry of the viscometer. According to the measurement, mixtures with above 25 μ m x₅₀ has shown Bingham-plastic behaviour with yield stress of 15 Pa, however, grinding has decreased the coefficient of rigidity [η , mPas]. If the fineness is higher the rheology changed and the mixture became Newtonian liquid. When the specific surface area of fly ash was 5876.8 cm²/g the viscosity has reached 60 mPas. Based on Fig. 3 it can be observed that the increase of specific surface area of fly ash decreased the viscosity of geopolymer paste.

Geopolymer foam strength and specimen density

In Fig. 4 the compressive strength and specimen density values of geopolymer foam can be seen as function of grinding fineness of fly ash. Based on this figure it can be stated that the low specimen density resulted in low compressive strength.

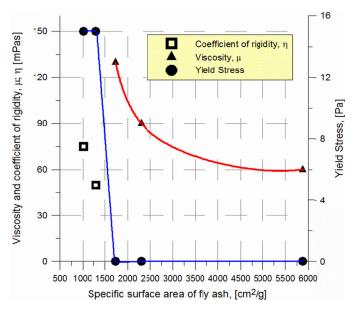


Fig. 3. Effect of grinding fineness on rheology of geopolymer pastes

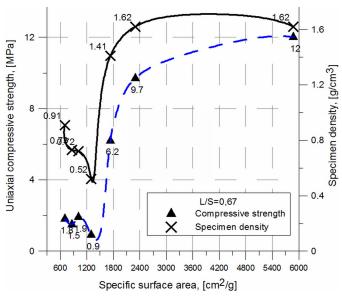


Fig. 4. Effect of grinding fineness of fly ash on geopolymer foam strength and density

When the specific surface area of fly ash was in the range of 700-1300 cm²/g the compressive strength value of geopolymer foam was relatively low (between 0.9 and 1.9 MPa). The lowest compressive strength was 0.9 MPa which belonged to 1300 cm²/g SSA of fly ash. In this case the specimen density of geopolymer foam was the lowest, with 0.52 g/cm³. Above ~1700 cm²/g SSA of fly ash the compressive strength and specimen density of geopolymer foam increased. The highest compressive strength (12 MPa) and specimen density (1.62 g/cm³) can be achieved at 5876.8 cm²/g SSA of fly ash ($x_{50} = 10.79 \mu m$).

Cell structure of geopolymer foam

The mean pore size of geopolymer foam can be seen in Fig. 5. The geopolymer foam had the highest mean pore size when the specific surface area of fly ash was reached $1300\,\mathrm{cm}^2/\mathrm{g}$

 $(x_{50} = 42.23 \mu m)$. In this case the mean pore size was 2.6 mm. When the specific surface area of fly ash was below 857 cm²/g or above 5800 cm²/g the mean pore size of geopolymer foam was below 0.7 mm.

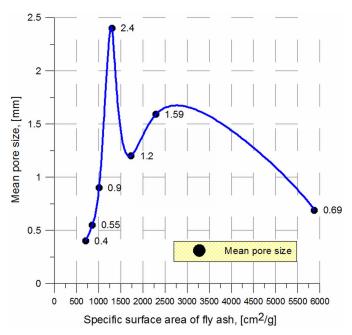


Fig. 5. Mean pore size of geopolymer foam

Based on these results it can be stated that the formed cell structure depends on the rheology of geopolymer paste. Due to the high viscosity of geopolymer paste the formation of pore was inhibited, the pore size of geopolymer foam were too small (few macropores are detected (Fig. 6A)). Due to too low viscosity the oxygen gas has gone easily from the geopolymer paste. As a result of this the geopolymer foam had compact structure with

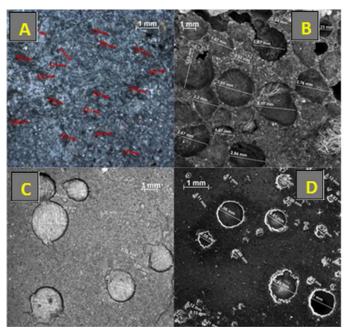


Fig. 6. Optical microscopy images of GPF (A. 0 min, B. 20 min, C. 60 min, D. 120 min grinding time of fly ash)

few bigger pores (macropores are below 2 mm (Fig. 6C and D)). The density increased up to 1.62 g/cm^3 .

Material structure

Based on the FTIR spectra of raw and mechanically activated fly ash (Fig. 7), significant absorbance can be observed at 1054 cm⁻¹, which is usually attributed to T-O-Si (T = Si or Al) bonds asymmetric stretching vibration, while at 546 cm⁻¹ can be assigned to Si-O-Si and Al-O-Si bonds symmetric stretching vibration.

Peak corresponding to 796 cm⁻¹ related to Si–O–Si bonds symmetric stretching vibration [12,13]. It can be stated, that the increasing grinding time resulted in slightly higher peak intensities, which can be associated to higher infra absorbance due to higher specific surface area of fine particles.

However, based on the FTIR spectra of geopolymer foams (Fig. 8) it can be observed that the grinding of fly ash caused changing of geopolymer foam material structure. Based on Panias et al [12] peak at 3318 cm⁻¹ corresponding to –OH, HOH bonds stretching vibration while the peak at 1641 cm⁻¹ sign the HOH bonds bending vibration. Peak at 1404 cm⁻¹ related to O–C–O stretching vibration. Peak observed at 1054 cm⁻¹ in the case of fly ash shifted lower wavenumber (992 cm⁻¹) after geopolymerization, which indicates structural reorganization.

It is associated with dissolution of amorphous phase of fly ash in strong alkali media according to Panias et al [12]. The peak intensities of gopolymer foam increased as function of increasing grinding time. It indicates higher rate of geopolymerisation according to Kumar et al [14].

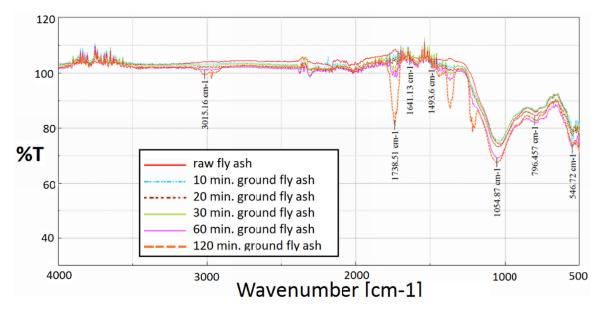


Fig. 7. FTIR spectra of mechanically activated fly ash

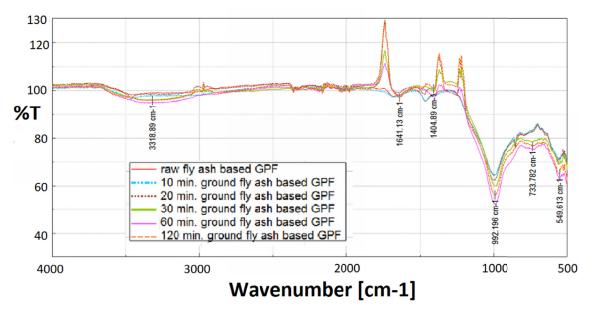


Fig. 8. FTIR spectra of geopolymer foam from mechanically activated fly ash

Conclusions

Based on the experimental results presented above the following conclusions can be drawn:

- The rheology of geopolymer paste can be changed by decrease of particle size (increase of specific surface area) of fly ash by keeping constant liquid/powder ratio (L/S).
- Beside constant liquid/powder ratio (L/S = 0.82) if the specific surface area of fly ash was below 1300 cm²/g (<x₅₀ = 25 μm), the geopolymer paste has shown Non-Newtonian (Bingham plastic) behaviour with yield stress of 15 Pa. The grinding has decreased the coefficient of rigidity [η, mPas]. If the fineness is higher the rheology changes and the mixture become Newtonian liquid.
- The formation of porous structure strongly influences on the rheology of geopolymer paste. If the viscosity of geopolymer paste was low, the geopolymer foam shown quite compact structure with few higher ($< x_{mean} = 2 \text{ mm}$) pores (Fig. 6C and D). In this case the generated oxygen gas had gone easily from geopolymer paste. The high viscosity resulted in foam structure with small pores ($x_{mean} = 0.4 \text{ mm}$) (Fig. 6A).
- The compressive strength of geopolymer foam depends on the pore structure. Based on the measurements, the highest porosity (0.75) resulted the lowest compressive strength (0.9 MPa).
- Optimal fly ash fineness (SSA = $1300 \text{ cm}^2/\text{g}$, $x_{50} = 42 \text{ μm}$) resulted in 0.9 MPa compressive strength and 0.52 g/cm³ density of geopolymer foam.

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REFERENCES

- [1] V. Vaou, D. Panias, Miner. Eng. 23 (14), 1146-1151 (2010).
- [2] J. Davidovits, Geopolymer chemistry and application, Saint-Quentin France, 2011
- [3] J. Davidovits, J. Materials Education 16, 91-138 (1994).
- [4] G. Mucsi, J. Lakatos, Z. Molnár, R. Szabó, in D. Cygas, T. Tollazzi (Ed.), Paper 39, Vilnius Gediminas Technical University Press, Vilnius (2014).
- [5] K. Komintsas, D. Zaharaki, Mineral Engineering 20, 1261-1277 (2007).
- [6] Z. Abdollahnejad, F. Pacheco-Torgal, T. Félix, W. Tahri, J. Barroso Aguiar, Construction and Building Materials 80, 18-30 (2015).
- [7] P. Hlaváček, V. Šmilauer, F. Škvára, L. Kopecký, R. Šulc, Journal of the European Ceramic Society 35(2), 703-709 (2015).
- [8] W.D.A. Rickard, A. van Riessen, Cement and Concrete Composites 48, 75-82 (2014).
- [9] J.G. Sanjayan, A. Nazari, L. Chen, G.H. Nguyen, Construction and Building Materials **79**, 236-244 (2015).
- [10] V. Ducman, L. Korat, Materials characterization 113, 207-213 (2016).
- [11] M.S. Cilla, P. Colombo, M.R. Morelli, Ceramics International 40, 5723-5730 (2014).
- [12] D. Panias, I.P. Giannopoulou, T. Perraki, Colloids and Surfaces A. 301, 246-54 (2007).
- [13] W.K.W. Lee, J.S.J. van Deventer, Colloids and Surfaces A. 211, 49-66 (2002).
- [14] S. Kumar, R. Kumar, Ceramics International 37, 533-541 (2011).
- [15] G. Mucsi, Á. Szenczi, Z. Molnár, J. Lakatos, Environmental Engineering and Landscape Management 24(1), 48-59 (2016).