

Małgorzata RUTKOWSKA-GORCZYCA*, Anita PTAK**, Marcin WINNICKI***

ANALYSIS OF THE TRIBOLOGICAL PROPERTIES OF Cu-ATiO₂ COMPOSITE COATINGS APPLIED BY THE COLD SPRAY METHOD

ANALIZA WŁAŚCIWOŚCI TRIBOLOGICZNYCH POWŁOK KOMPOZYTOWYCH Cu-ATiO₂ NANOSZONYCH METODĄ COLD SPRAY

Key words: composite coatings, LPCS, friction, abrasive wear.

Abstract: The properties of copper have been known and used for a very long time, and research has also been carried out for a long time to expand the applications of this material. One of the methods increasing the bactericidal and bacteriostatic effect of copper is modification by means of the TiO₂ phase. The research was conducted in order to determine the impact of modification of copper coatings with TiO₂ titanium dioxide on their tribological properties. The paper presents the results of studies on tribological wear of composite coatings applied on steel using the method of low-pressure cold gas spraying (LPCS). The tests of resistance to abrasive wear were carried out in a ball-disc combination in reciprocating motion. The analysis of the resistance to abrasive wear of the tested coatings included the determination of the impact of the pressure force on the intensity of wear and the kinetic friction coefficient of the tested friction pairs. It was found that the samples covered only with copper coatings were characterized by a higher value of friction coefficient in relation to the substrate made of AISI 316L steel. The modification of copper with the submicron particles TiO₂ fraction does not increase the value of friction coefficient. The value of this parameter is maintained at a similar level regardless of the applied counterspecimen.

Słowa kluczowe: powłoki kompozytowe, LPCS, tarcie, zużycie ściernie.

Streszczenie: Właściwości miedzi są znane i wykorzystywane od bardzo dawna, od dawna również są prowadzone badania nad zwiększeniem możliwości zastosowania tego materiału. Jedną z metod zwiększających działanie bakteriobójcze i bakteriostatyczne miedzi jest modyfikacja fazą TiO₂. Badania przeprowadzono w celu określenia wpływu modyfikacji powłok miedzianych ditlenkiem tytanu TiO₂ na ich właściwości tribologiczne. W pracy zaprezentowano wyniki badań zużycia tribologicznego powłok kompozytowych nanoszonych na stal metodą niskociśnieniowego natrysku zimnym gazem (LPCS). Badania odporności na zużycie ściernie wykonano w skojarzeniu kula–tarcza, w ruchu posuwisto-zwrotnym. Analiza odporności na zużycie ściernie badanych powłok obejmowała określenie wpływu siły nacisku na intensywność zużywania i współczynnik tarcia kinetycznego badanych par trących. Stwierdzono, że próbki pokryte wyłącznie powłoką miedzianą charakteryzują się wyższą wartością współczynnika tarcia, w stosunku do podłoża ze stali AISI 316L. Modyfikacja miedzi frakcją submikronowych cząstek TiO₂ nie powoduje zwiększenia wartości współczynnika tarcia. Wartość tego parametru zostaje zachowana na podobnym poziomie niezależnie od zastosowanego obciążenia przeciwpróbki.

INTRODUCTION

The properties of copper and its alloys have been used for centuries in many areas of life and science. In addition to

its aesthetic qualities, copper has a number of physico-chemical properties that allow ever new applications. New technologies and scientific discoveries open almost unlimited application possibilities [L. 1–4].

* ORCID: 0000-0003-2712-5914. Wrocław University of Science and Technology, Faculty of Mechanical Engineering, Department of Vehicle Engineering, Łukasiewicza 5 Street, 50-370 Wrocław, Poland.

** ORCID: 0000-0002-2177-5812. Wrocław University of Science and Technology, Faculty of Mechanical Engineering, Department of Fundamentals of Machine Design and Mechatronic Systems, Łukasiewicza 5 Street, 50-370 Wrocław, Poland.

*** ORCID: 0000-0003-2766-116X. Wrocław University of Science and Technology, Faculty of Mechanical Engineering, Department of Metal Forming, Welding and Metrology, Łukasiewicza 5 Street, 50-370 Wrocław, Poland.

The antimicrobial activity of copper and copper-based alloys is also well documented, and copper has recently been registered in the USA with the Environmental Protection Agency as the first solid antimicrobial material [L. 1–4]. As a result of clinical studies, it was found that copper can be used on touchable surfaces such as door handles, bathroom fittings or bed rails to reduce nosocomial infections [L. 5]. At a time when there is a global epidemic, caused by a huge number of cases and deaths due to COVID-19, this issue is particularly relevant. The global situation makes it necessary to extend these issues also to coatings on elements for everyday use. At present, the elements on which micro-organisms persist are those with which we are in contact in public places (shops, parks, entertainment centres); they constitute the greatest threat and on them the COVID-19 virus takes the greatest toll.

Literature reports that copper-based composite coatings offer interesting results and possibilities by, e.g., increasing microbiological activity [L. 6–16]. To diversify and improve the microbiological activity of copper, both TiO_2 and Ag are used, either in the form of an interlayer or modifier. The addition of active ceramics is intended to increase the bactericidal activity of the surface, and the addition of TiO_2 phase also uses photocatalytic activity [L. 17–22].

The method of low-pressure cold gas spraying (LPCS) allows to apply both soft metals (Cu, Al, Zn, Sn) using the mechanism of plastic deformation of powder particles, as well as modification of the base powder with ceramic particles (SiC , Al_2O_3 , TiO_2) [L. 20, 21, 23]. Usually, the presence of a hard ceramic fraction causes strengthening of the coating and the reduction of porosity of the soft metallic matrix. It is different in case of the amorphous form of aTiO_2 , which, due to its structure, does not cause a stronger degree of crushing of metallic matrix [L. 24–26]. Low temperature (400°C) during the LPCS process does not cause phase changes in aTiO_2 , leaving the ceramic phase aTiO_2 in amorphous form. Scientific research [L. 6, 7, 18] confirms that this form of aTiO_2 also shows biological and physicochemical activity, causing changes in the properties of the base material, e.g., Co and Cu. Our previous research [L. 12] has shown that using LPCS methods makes it possible to produce durable Cu- TiO_2 composite coatings. Additionally, on the basis of our own research, it was found that aTiO_2 modification results in the shift of corrosion potentials towards more positive values, which means an increase in corrosion resistance of Cu- aTiO_2 composite coatings. However, there are no reports in the literature about testing mechanical properties of this type of composite coatings. Therefore, the study presented in this paper was aimed at determining the impact of modification of copper coatings with a fraction of amorphous titanium dioxide on tribological properties of coatings applied by the Cold Spray method. Bioactive coatings for applications

on public utility surfaces (handles, handrails, and others) should be durable, have appropriate thicknesses, and be resistant to frequent touching and friction. Therefore, relevant tests presented in this publication have been undertaken.

MATERIALS AND MEASURING METHODS

Tribological studies were carried out on copper coatings and copper coatings modified with wt10% of amorphous titanium dioxide. Commercially available copper powder of dendritic morphology and fraction size from $10\ \mu\text{m}$ to $50\ \mu\text{m}$ was used. Amorphous submicron titanium dioxide aTiO_2 , produced using sol-gel technology, with submicron granulation in the range of $400\text{--}600\ \text{nm}$, which forms agglomerates of $10\text{--}70\ \mu\text{m}$, was used for the modification [L. 25]. The mass fraction of titanium dioxide was selected on the basis of preliminary tests of coating microstructure. Powders were mixed with the use of laboratory mill for $t = 60\ \text{min}$. Copper powders and Cu- aTiO_2 composite powders were applied on an AISI 316L austenitic steel substrate, $10 \times 70\ \text{mm}$ in size and $1\ \text{mm}$ thick. Before spraying, the substrate was blasted at a pressure of $p = 0.6\ \text{MPa}$ and using electrocorundum of $0.8\ \text{mm}$ granulation. Low-pressure cold gas spraying (LPCS) was used to apply the coatings at the technological parameters presented in **Table 1** (determined by means of an experiment). The porosity was determined using a graphical semi-quantitative analysis using the ImageJ software. Five images with a magnification of $2500\times$ were selected, and the average value from the measurements was taken as the result. The coating thickness after application was $600\ \mu\text{m}$ for the copper coating and $200\ \mu\text{m}$ for the composite coating (due to different particle sizes of Cu and aTiO_2 , the deposition of coatings is more difficult). An analysis of the surface topography of coatings after the mechanical polishing process (using 1 and $6\ \mu\text{m}$ polishing slurries) was carried out before and after tribological tests in order to evaluate changes in coatings.

Table 1. Technological parameters of coating application using the LPCS method

Tabela 1. Parametry technologiczne nanoszenia powłok metodą LPCS

Powder	Gas heating temperature	Gas pressure	Sliding speed
Cu, Cu- aTiO_2	400°C	$0.9\ \text{MPa}$	$10\ \text{mm/s}$

Friction tests were carried out on a ball-on-plate test stand which enables one to carry out sliding friction tests in a reciprocating motion [L. 27]. A silicon carbide SiC ball with a diameter $d = 3.969\ \text{mm}$ was used for testing. The counterspecimen was a steel plate made of AISI 316L steel with an applied coating.

During the tests, the ball was pressed against the counterspecimen by F_n with three different values: $F_n = 5$ N, $F_n = 10$ N and $F_n = 15$ N. A steel plate with an applied coating was fixed on a movable carriage. Three series of measurements were carried out in the test, and, in each of them, the plate mounted on a moving trolley performed a series of 200 movement cycles. The movement cycle consisted of two movements in both directions, achieving a maximum relative speed $v_{max} = 5$ mm/s with a stroke of $S = 10$ mm. The motion time in each direction was $t = 0.4$ s. During the measurements, the friction force values were recorded at a frequency of $f = 100$ Hz, and the measurement limit error determined based on the sensor and amplifier class was $\Delta gr = 0.05$ N. The tests were conducted under technically dry friction conditions. The obtained measurement results were subject to statistical calculations.

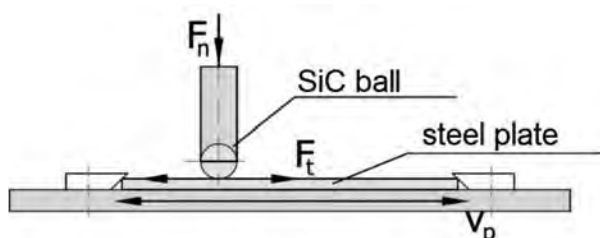


Fig. 1. Diagram of the test stand for alternating motion using a ball-on-plate combination [L. 28]

Rys. 1. Schemat stanowiska do badań w ruchu przemiennym z wykorzystaniem skojarzenia typu ball-on-plate [L. 28]

Five hardness measurements were carried out for each coating material, in accordance with the PN-EN ISO 6507-1:2007 standard, using the Vickers method and a Leco LM-248AT hardness meter. The measurements were carried out at a load of $m = 1$ kG (9.81 N). Analyses of the coating surface after friction with the use of scanning electron microscopy methods were carried out with the use of a SEM Phenom G2 microscope, using a BSE detector.

RESULTS AND DISCUSSION

Coatings obtained in the LPCS process show a compact and homogeneous microstructure in terms of morphology but are characterized by different degrees of plastic deformation. Thus, there is a different degree of deformation and strengthening of individual coating particles. The different structure has a significant impact on the hardness of coatings (**Table 2**). An averaged five measurements of hardness of coatings, determined in the cross-section, for a copper coating, show a hardness of 86 HV1. Modification with the amorphous ceramic phase TiO_2 causes a decrease in hardness to 63 HV1.

Table 2. Averaged measurements of hardness of applied coatings

Tabela 2. Uśrednione pomiary twardości naniesionych powłok ($n = 5$)

Coating	HV1	Deviation
Cu	86	8.7
Cu-aTiO ₂	63	6.5

The analysis of the surface of samples with Cu coatings showed a small degree of porosity. A lower density of Cu coatings is visible in areas that were not matched entirely in highly developed dendritic copper powder particles (**Figs 2a and b**). In Cu-aTiO₂ composite coatings, potential porosity areas are filled with the ceramic phase aTiO₂. The aTiO₂ inclusions are visible in the form of black inclusions, clearly accumulated within the inter-dendritic spaces of the copper powder (**Fig. 3a**). In case of composite coatings, porosity is visible within the presence of aTiO₂ particles themselves (**Fig. 3b**). It is connected with amorphous structure and lack of plastic deformation of aTiO₂ particles during LPCS application. Due to high ductility, dendritic copper particles, as a result of collision with the substrate, are strongly deformed and fill the space between the particles tightly. Cu coatings show a porosity of 1–2 %. In the case of composite coatings, submicron aTiO₂ particles are unevenly distributed in the inter-dendritic spaces of copper powder. In composite coatings, uneven filling of spaces between metallic matrix particles is visible, i.e. they show higher porosity in relation to copper coatings at the level of 3–4%.

The average value of the kinetic friction coefficient in the combination SiC/Cu coating and the lowest load $F_n = 5$ N was $\mu = 0.33$. Further studies have shown that the addition of 10% aTiO₂ causes this coefficient to remain at a similar level (**Fig. 4**). In the case of a copper coating, a decrease in this coefficient was observed with an increase in the loading force from $\mu = 0.35$ to $\mu = 0.28$. The error bar values allow us to conclude that the most stable measurement conditions can be obtained for the tested pair with the lowest loading force $F_n = 5$ N. With higher loads, i.e. $F_n = 10$ N and $F_n = 15$ N, the spread of the results even reaches approx. 30% of the values obtained.

For the analysis of wear, a method of measuring the width of the friction traces was used to supplement the qualitative assessment of wear [**L. 28–35**]. The analysis of the friction traces produced by the friction processes clearly indicates that the highest wear is observed at the highest load applied, i.e. $F_n = 15$ N (**Fig. 5**). In this range, the width of the friction trace takes the same values, at $x = 400$ μm . This shows that a 10% share of the modifying phase under the specified loads does not affect the tribological properties. For smaller load values, the reduced width of the friction traces is obtained for the modifying sample and the load $F_n = 5$ N, the reduction of

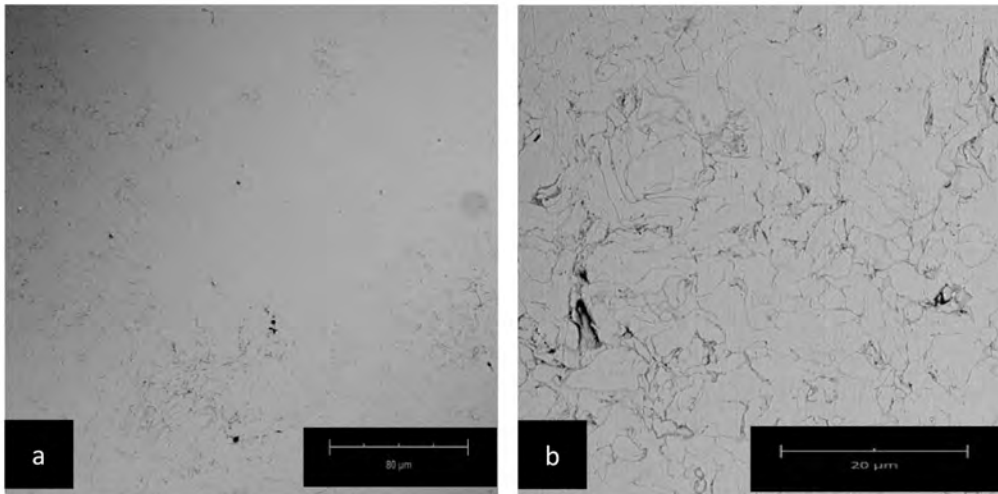


Fig. 2. Cu coating surface, before tribological tests (SEM): a) morphology of the Cu coating, b) enlarged fragment from the area a) with boundaries between the grains of the Cu powder

Rys. 2. Powierzchnia powłoki miedzianej Cu przed testami tribologicznymi (SEM): a) morfologia powłoki Cu z niewielkim stopniem porowatości, b) powiększony fragment z obszaru a) z widocznymi granicami pomiędzy ziarnami proszku Cu

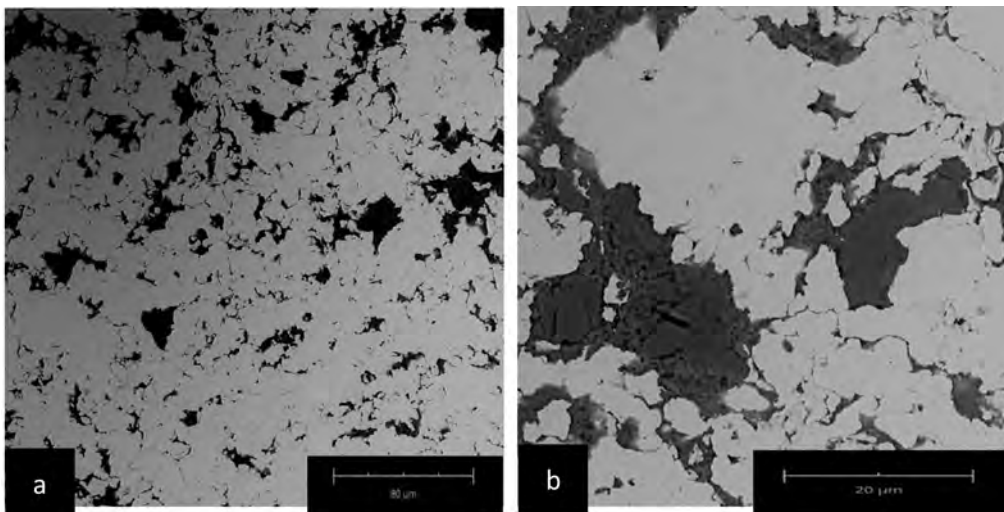


Fig. 3. Surface of Cu-aTiO₂ composite coatings, before tribological tests (SEM): a) morphology of the Cu-aTiO₂ coating, b) enlarged fragment from area a)

Rys. 3. Powierzchnia kompozytowych powłok Cu-aTiO₂ przed testami tribologicznymi (SEM): a) morfologia powłoki Cu (jasny obszar) – aTiO₂ (ciemny obszar), b) powiększony fragment z obszaru a)

the friction trace width from $x = 280 \mu\text{m}$ to $x = 240 \mu\text{m}$. For both loads $F_n = 5 \text{ N}$ and $F_n = 10 \text{ N}$ in the case of the copper coating, no changes in the width of the friction trace were observed; they remain at a similar level of $x = 280 \mu\text{m}$. A reduction in the width of the friction trace at lower loads indicates greater resistance to the abrasion of composite coatings. The wear of the silicon ball at the given parameters of the experiment was not measurable.

An analysis of the topography of the coatings surface after tribological tests was also carried out using scanning electron microscopy methods. The tests showed that, as a result of contact between the coatings with the ball, the coatings become worn evenly. The nature of wear

was determined as plastic, and in no case were any brittle splinters of coatings observed. Both copper coatings Cu (**Fig. 6**) and composite coatings Cu-aTiO₂ (**Fig. 7**) show a similar mechanism of wear without the occurrence of areas of brittle cracks, and the addition of the modification phase does not change this mechanism. This is also confirmed by the friction coefficient values recorded for both types of coatings. Modification with amorphous aTiO₂ fraction, as opposed to crystalline particles, does not cause any cracking in the coating [**L. 19, 28, 32–35**]. The use of amorphous titanium dioxide powder, due to its submicron granulation and additionally an amorphous form, does not lead to brittle cracking.

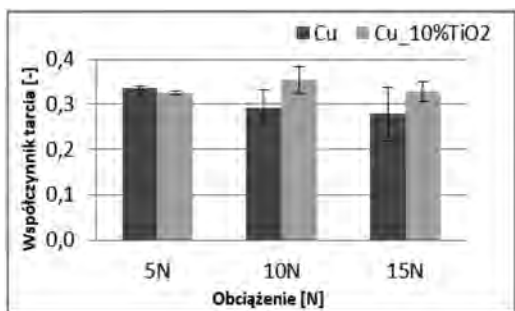


Fig. 4. Averaged friction coefficient values for copper coatings and Cu-aTiO₂ composite coatings for various values of loading force

Rys. 4. Uśrednione wartości współczynnika tarcia kinetycznego w badanych skojarzeniach dla różnych wartości siły obciążającej

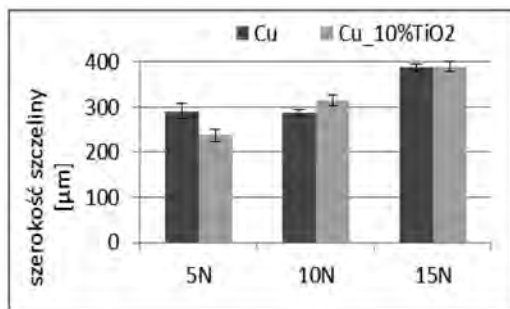


Fig. 5. Average values of the friction path width after friction tests, for Cu coatings and Cu-aTiO₂ composite coatings, for different values of loading force

Rys. 5. Uśrednione wartości szerokości śladu tarcia po próbach tarcia, dla powłok miedzianych oraz kompozytowych powłok Cu-aTiO₂ dla różnych wartości siły obciążającej

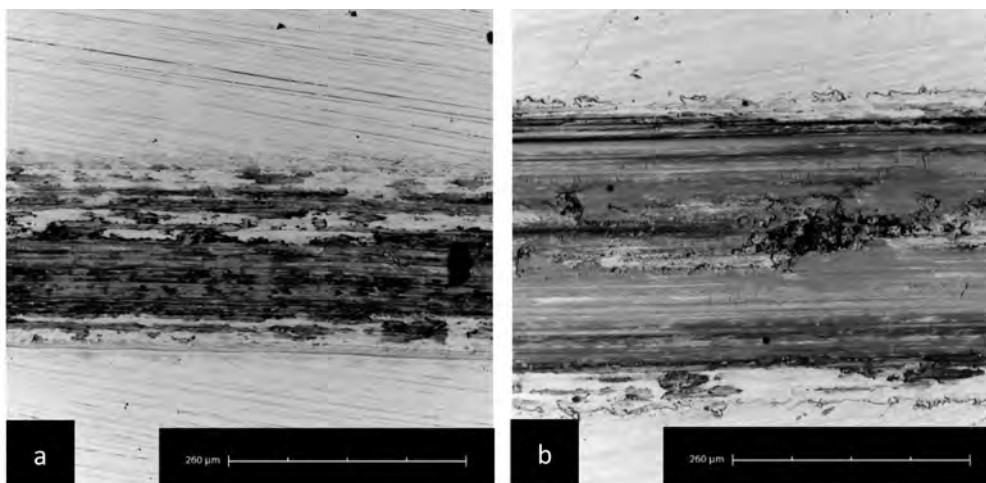


Fig. 6. Examples of wear and tear formed after tribological tests carried out on a Cu coating under load (SEM): a) $F_n = 5\text{ N}$, b) $F_n = 15\text{ N}$

Rys. 6. Przykładowe ślady zużycia powstałe po badaniach tribologicznych przeprowadzonych na powłoce miedzianej Cu przy obciążeniu (SEM): a) $F_n = 5\text{ N}$, b) $F_n = 15\text{ N}$

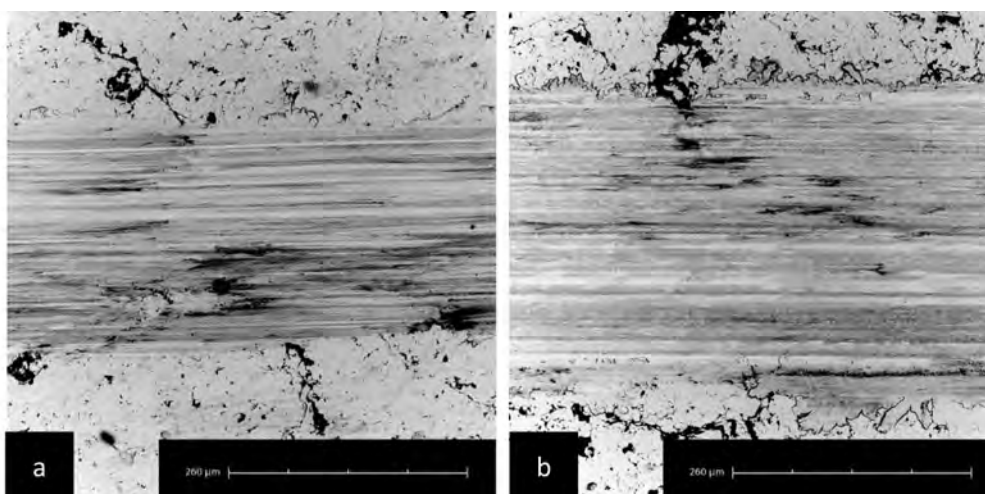


Fig. 7. Examples of wear and tear formed after tribological tests carried out on composite Cu-aTiO₂ coating under load (SEM): a) $F_n = 5\text{ N}$, b) $F_n = 15\text{ N}$

Rys. 7. Przykładowe ślady zużycia powstałe po badaniach tribologicznych przeprowadzonych na kompozytowej powłoce Cu-aTiO₂ przy obciążeniu (SEM): a) $F_n = 5\text{ N}$, b) $F_n = 15\text{ N}$

CONCLUSIONS

The tribological and instrumental studies lead to the following conclusions:

- The addition of a ceramic phase in the form of 10% by weight of amorphous aTiO₂ reduces the hardness of coatings. The recorded decrease of hardness value is 86 HV1 for copper coating and 63 HV1 for Cu-aTiO₂ composite coating.
- The change of load (5, 10, 15 N) does not affect the friction coefficient in combination with the composite coating.
- For the Cu-aTiO₂ composite coating, the value of the friction coefficient is maintained at a similar level regardless of the load value, taking into account the error bars obtained.
- Observation of the wear marks on the coatings as a result of the friction process showed that both copper coatings and Cu-aTiO₂ composite coatings show a high degree of plasticity. Observation of images recorded with the use of scanning electron microscopy methods showed that there are no brittle changes in coatings.
- The research shows that modification of the copper coating with a 10% share of titanium dioxide does not trigger the deterioration of tribological properties of coatings. This allows the extension of the potential application of composite coatings to biologically active coatings used on elements of public use (handles, handrails, and others).

REFERENCES

1. Park, Y.J.; Song, Y.H.; An, J.H.; Song, H.J.; Anusavice, K.J.: Cytocompatibility of pure metals and experimental binary titanium alloys for implant materials. *J. Dent.* 2013, 41, pp. 1251–1258.
2. Wilks, S.A.; Michels, H.; Keevil, C.W.: The survival of *Escherichia coli* O157 on a range of metal surfaces. *Int. J. Food Microbiol.* 2005, 105, pp. 445–454.
3. Wojcieszak, D.; Mazur, M.; Kalisz, M.; Grobelny, M.: Influence of Cu, Au and Ag on structural and surface properties of bioactive coatings based on titanium. *Mater. Sci. Eng. C* 2017, 71, pp. 1115–1121.
4. Grass, G.; Rensing, C.; Solioz, M. Metallic copper as an antimicrobial surface. *Appl. Environ. Microbiol.* 2011, 77, pp. 1541–1547.
5. Chen, S.; Guo, Y.; Zhong, H.; Chen, S.; Li, J.; Ge, Z.; Tang, J.: Synergistic antibacterial mechanism and coating application of copper/titanium dioxide nanoparticles. *Chem. Eng. J.* 2014, 256, pp. 238–246.
6. El-Eskandrany, M.S.; Al-Azmi, A.: Potential applications of cold sprayed Cu50 Ti20 Ni30 metallic glassy alloy powders for antibacterial protective coating in medical and food sectors. *J. Mech. Behav. Biomed. Mater.* 2016, 56, pp. 183–194.
7. Yamada, M.; Isago, H.; Nakano, H.; Fukumoto, M.: Cold spraying of TiO₂ photocatalyst coating with nitrogen process gas. *J. Therm. Spray Technol.* 2010, 19, pp. 1218–1223.
8. Huang, T.; Sui, M.; Li, J.: Inactivation of *E. coli* by nano-Cu/MWCNTs combined with hydrogen peroxide. *Sci. Total Environ.* 2017, 574, pp. 818–828.
9. Mohajeri, S.; Dolati, A.; Ghorbani, M.: The influence of pulse plating parameters on the electrocodeposition of Ni-TiO₂ nanocomposite single layer and multilayer structures on copper substrates. *Surf. Coatings Technol.* 2015, 262, pp. 173–183.
10. Ramalingam, S.; Muralidharan, V.S.; Subramania, A.: Electrodeposition and characterization of Cu-TiO₂ nanocomposite coatings. In *Proceedings of the Journal of Solid State Electrochemistry; 2009; Vol. 13, pp. 1777–1783.*
11. Rutkowska-Gorczyca, M.: X-ray diffraction and microstructural analysis of Cu-TiO₂ layers deposited by cold spray. *Mater. Sci. Technol. (United Kingdom)* 2020.
12. Sanjabi, S.; Shirani, A.: The morphology and corrosion resistance of electrodeposited Co-TiO₂ nanocomposite coatings. *Mater. Corros.* 2011, p. 63.
13. Vilardell, A.M.; Cinca, N.; Dosta, S.; Cano, I.G.; Guilemany, J.M.: Feasibility of using low pressure cold gas spray for the spraying of thick ceramic hydroxyapatite coatings. *Int. J. Appl. Ceram. Technol.* 2019, 16, pp. 221–229.

14. Winnicki, M.; Baszczuk, A.; Jasiorski, M.; Borak, B.; Małachowska, A.: Preliminary studies of TiO₂ nanopowder deposition onto metallic substrate by low pressure cold spraying. *Surf. Coatings Technol.* 2019, 371, pp. 194–202.
15. Wojcieszak, D.; Kaczmarek, D.; Antosiak, A.; Mazur, M.; Rybak, Z.; Rusak, A.; Osekowska, M.; Poniedziałek, A.; Gaman, A.; Szponar, B.: Influence of Cu-Ti thin film surface properties on antimicrobial activity and viability of living cells. *Mater. Sci. Eng. C* 2015, 56, pp. 48–56.
16. Chung, C.J.; Chiang, C.C.; Chen, C.H.; Hsiao, C.H.; Lin, H.I.; Hsieh, P.Y.; He, J.L.: Photocatalytic TiO₂ on copper alloy for antimicrobial purposes. *Appl. Catal. B Environ.* 2008, 85, pp. 103–108.
17. Mungkalasiri, J.; Bedel, L.; Emieux, F.; Doré, J.; Renaud, F.N.R.; Maury, F.: DLI-CVD of TiO₂-Cu antibacterial thin films: Growth and characterization. *Surf. Coatings Technol.* 2009, 204, pp. 887–892.
18. Utu, I.D.; Marginean, G.; Hulka, I.; Serban, V.A.; Cristea, D.: Properties of the thermally sprayed Al₂O₃-TiO₂ coatings deposited on titanium substrate. *Int. J. Refract. Met. Hard Mater.* 2015, 51, pp. 118–123.
19. Winnicki, M.; Rutkowska-Gorczyca, M.; Małachowska, A.; Piwowarczyk, T.; Ambroziak, A.: Microstructure and Corrosion Resistance of Aluminium and Copper Composite Coatings Deposited by LPCS Method. *Arch. Metall. Mater.* 2016, 61, pp. 1945–1952.
20. Winnicki, M.; Baszczuk, A.; Rutkowska-Gorczyca, M.; Ambroziak, A.: Corrosion resistance of tin coatings deposited by cold spraying. *Taylor Fr.* 2016, 32, pp. 691–700.
21. Zheng, X.; Shi-Peng, S.; Cheng, C.; Shi, L.; Cheng, R.; Dong-Hai, D.: Photocatalytic disinfection performance in virus and virus/bacteria system by Cu-TiO₂ nanofibers under visible light. *Environ. Pollut.* 2018, 237, pp. 452–459.
22. AlMangour, B.: Fundamentals of cold spray processing: Evolution and future perspectives. In *Cold-Spray Coatings: Recent Trends and Future perspectives*; Springer International Publishing, 2017; pp. 3–24.
23. Vargas, M.A.; Rodríguez-Páez, J.E.: Amorphous TiO₂ nanoparticles: Synthesis and antibacterial capacity. *J. Non. Cryst. Solids* 2017, 459, pp. 192–205.
24. Baszczuk, A.; Jasiorski, M.; Winnicki, M.: Low-Temperature Transformation of Amorphous Sol-Gel TiO₂ Powder to Anatase During Cold Spray Deposition. *J. Therm. Spray Technol.* 2018, 27, pp. 1551–1562.
25. Kaur, K.; Singh, C.V. Amorphous TiO₂ as a photocatalyst for hydrogen production: a DFT study of structural and electronic properties. *Energy Procedia* 2012, 29, 291–299.
26. Kowalewski, P.; Leśniewski, T.; Wieleba, W.: Stanowisko do badań tribologicznych w złożonym ruchu cyklicznym toczno-ślizgowym. *Tribologia* 2007, pp. 303–311.
27. Łęcka, K.M.; Antończak, A.J.; Kowalewski, P.; Trzeźniński, M.: Wear resistance of laser-induced annealing of AISI 316 (EN 1.4401) stainless steel. *Laser Phys.* 2018, 28, pp. 1–8.
28. Berman, D.; Erdemir, A.; Sumant, A.V.: Few layer graphene to reduce wear and friction on sliding steel surfaces. *Carbon N. Y.* 2013, 54, pp. 454–459.
29. Fernández, E.; Cadenas, M.; González, R.; Navas, C.; Fernández, R.; Damborenea, J. De Wear behaviour of laser clad NiCrBSi coating. *Wear* 2005, 259, pp. 870–875.
30. Haseeb, A.S.M.A.; Albers, U.; Bade, K.: Friction and wear characteristics of electrodeposited nanocrystalline nickel-tungsten alloy films. *Wear* 2008, 264, pp. 106–112.
31. Bolelli, G.; Cannillo, V.; Lusvardi, L.; Manfredini, T.: Wear behaviour of thermally sprayed ceramic oxide coatings. *Wear* 2006, 261, pp. 1298–1315.
32. Chao, W.L.; Wan, Y.; Wang, Y.X.; Liu, C.S.: Tribological properties of Cu-doped TiO₂ films. *Wuli Huaxue Xuebao/Acta Phys. – Chim. Sin.* 2010, 26, pp. 2317–2322.
33. Ramesh, C.S.; Ahmed, R.N.; Mujeebu, M.A.; Abdullah, M.: Fabrication and study on tribological characteristics of cast copper-TiO₂-boric acid hybrid composites. *Mater. Des.* 2009, 30, pp. 1632–1637.
34. Vencl, A.; Rajković, V.; Zivic, F.; Mitrović, S.; Cvijović-Alagić, I.; Jovanovic, M.T.: The effect of processing techniques on microstructural and tribological properties of copper-based alloys. *Appl. Surf. Sci.* 2013, 280, pp. 646–654.