Owing to its beneficial technological and mechanical properties, cast iron not only continues being the most widely used cast alloy but is also the material under constant development. Cast iron is used in nearly every branch of the industry. Because of the growing requirements from users of this material, new grades of cast iron of better mechanical and performance properties are searched. One of the lines in this search is connected with introduction to cast iron of alloying elements, which change the type of matrix and the morphological characteristics of graphite, or promote the crystallisation of carbides. The element of very interesting, though still not fully explored, properties is vanadium. Literature on cast iron with high vanadium content is also very scarce. In [1, 2], information on the structure of Fe-12.9% V-2.94% C alloys was given. In these alloys, the presence of fibrous $\gamma + VC_{1-x}$ eutectic was observed with volume content of vanadium carbide amounting to about 20%. The results of microstructural examinations of Fe-C-V alloys with carbon content in the range of 1.38÷4.16% and vanadium content in the range of 6.7÷15.5% as well as information on mechanical and tribological properties of these alloys are given in [3, 4]. The main source of information on the interaction of alloying constituents and their distribution in individual phases forming a system are phase equilibrium diagrams. However, the degree of difficulty and the cost of making such diagrams increase drastically with the increasing number of constituents included in the system [5]. There are several items in literature that describe the Fe-C-V phase equilibrium diagrams [6, 7, 8, 9], but they are not fully consistent with the metallographic practice. Hence follows the objective of the present studies involving manufacture of the Fe-V-C alloys characterised by different chemical composition with subsequent analysis of their microstructure. The paper focusses on alloys with near-eutectic chemical composition.

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2. METHODOLOGY OF RESEARCH

To conduct the planned investigations, a series of melts with variable content of carbon and vanadium was made in a Balzers type vacuum furnace in the atmosphere of argon. The following materials were used: Fe-V master alloys containing 81.7% vanadium, armco iron, and technically pure graphite. Moulds made from molochite flour based on sodium silicate hardened with CO₂ were preheated up to a temperature of 550 °C, and poured next with molten iron at a temperature of 1600 °C. From the knocked out castings, specimens for metallographic examinations were cut out.

Unetched specimens were examined under LEICA MEF4 M optical microscope and JEOL 5500LV scanning microscope with secondary electrons. This enabled differentiating between vanadium carbides and other phases, which was not always possible when optical microscope was the only tool used in investigations. For more detailed examination of the geometry of individual phases, the specimens were deep etched with aqua regia and examined under scanning microscope. The percent content of structural constituents was determined on a LEICA QWin automatic image analyser.

![Image](image.png)

*Fig. 1. Microstructure of near-eutectic Fe-C-V alloy after deep etching with aqua regia*

3. DISCUSSION OF RESULTS

Table 1 compares the chemical composition of the examined alloys, their microstructure and content of structural constituents. Besides vanadium eutectic, depending on $C/V$ ratio in cast iron (where: $C$, $V$ – carbon and vanadium content in alloy in %, respectively), the following microstructural constituents were observed to occur:
- alloyed ferrite (Fig. 2a), when the value of $C/V$ was 0.10±0.13,
- granular pearlite (Fig. 2b), when the value of $C/V$ was 0.15,
- lamellar pearlite (Fig. 2c), when the value of $C/V$ was 0.24±0.30.
Deep etching with aqua regia and observations under scanning microscope revealed
in the grains of regular eutectic (Figs. 1 and 3a), a continuous, ramified skeleton of va-
nadium carbide growing out from one common centre (Fig. 3b, c). It is also worth noting
that, at the final stage of growth, some fibres have been transformed into lamellar forms
(Fig. 3a, b).

A relationship between the content of carbon and vanadium corresponding to a near-
eutectic structure (eutectic line, Fig. 4a) is described with the following equation of re-
gression:

\[ C_e = 7.618 \cdot V_e^{-0.617} \]  

(1)

where: \( C_e \) and \( V_e \) – carbon and vanadium content in eutectic in \%, respectively.

The coefficient of correlation in this equation assumes a very high value of \( R = 0.99 \).

The degree of eutectic saturation \( S_e \) is a quotient of carbon content in alloy \( C \) divided
by the carbon content in eutectic \( C_e \). Considering the above and equation (1), the degree of
eutectic saturation in Fe-C-V alloys can be expressed with the following formula:

\[ S_e = \frac{C}{C_e} = \frac{C}{7.618 \cdot V_e^{-0.617}} \]  

(2)

For \( S_e < 1 \), \( S_e = 1 \) and \( S_e > 1 \) the alloys of, respectively, hypoeutectic, eutectic and
hypereutectic composition are obtained.
Fig. 2. Microstructure of near-eutectic Fe-C-V alloy
Fig. 3. Examples of the ramified vanadium carbide present in eutectic grains
Figure 4. The run of eutectic line in Fe-C-V alloys (a) and (b) projection of eutectic line (equation (1)) against the background of Fe-C-V system described in [6].

Figure 5. Effect of vanadium content on the content of eutectic carbides ($f_w$) in near-eutectic alloys.

Figure 5 shows the effect of vanadium and eutectic saturation degree $S_c$ on the content of vanadium carbides $f_w$, wherefrom it follows that increasing content of vanadium in near-eutectic alloys increases the content of vanadium carbide eutectic.
Figure 4b shows eutectic line plotted as a result of the carried out research, visible against the background of Fe-C-V alloy phase equilibrium diagram [6]. From this drawing it follows that the experimentally plotted eutectic line is shifted to the left in respect of the eutectic line in phase equilibrium diagram.

An example of the cooling curve for high-vanadium eutectic cast iron and a differential curve are shown in Figure 6.

![Cooling curve and its derivative plotted for eutectic high-vanadium cast iron](image)

**Fig. 6. Cooling curve and its derivative plotted for eutectic high-vanadium cast iron**

4. CONCLUSIONS

In near-eutectic alloys containing from 1.45 to 2.23% C and from 7.33 to 15.08% V, depending on the value of the $C/V$ ratio, besides vanadium eutectic, the following microstructural constituents are present:

- alloyed ferrite, when the value of $C/V$ is 0.10±0.13,
- granular pearlite, when the value of $C/V$ is 0.15,
- lamellar pearlite, when the value of $C/V$ is 0.24±0.30.

The derived relationship showing the eutectic saturation ratio enables easy determination if the examined Fe-C-V alloy is of hypoeutectic, eutectic or hypereutectic type, using only the results of chemical analysis, without the need for troublesome and time consuming metallographic examinations.

The eutectic line determined experimentally is shifted to the left respective of the eutectic line present in phase equilibrium diagram.

Increasing the content of vanadium in near-eutectic alloys increases the content of vanadium carbide eutectic.

**Acknowledgements**

*The work was supported by Project No. 10.10.170.396*
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Received
January 2011